

Scalable Conformal Electronics Based on Roll-to-Roll Exfoliated van der Waals Semiconductors

Yigit Sozen,* Esteban Zamora-Amo, Juan J. Riquelme, and Andres Castellanos-Gomez*



Cite This: *ACS Nano* 2026, 20, 17584–17596



Read Online

ACCESS |



Metrics & More



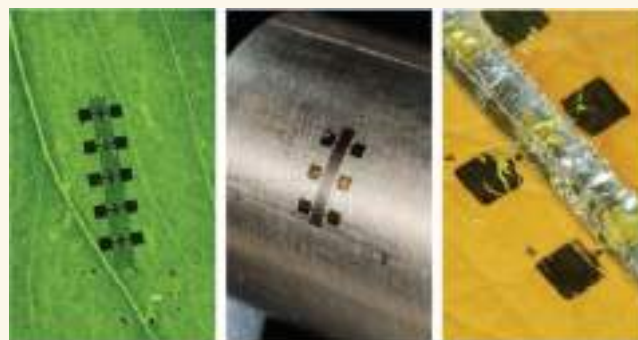
Article Recommendations



Supporting Information

ABSTRACT: Integrating electronic devices onto surfaces with complex topography such as skin, textiles, and biological tissues requires fabrication strategies that combine mechanical conformability with high electronic performance and scalable manufacturing. While two-dimensional (2D) semiconductors are promising materials for such applications, their integration into conformal electronic systems remains challenging because scalable liquid-phase processing typically yields films with limited electronic performance, whereas high-quality CVD materials require complex synthesis and transfer processes. Here, we establish a scalable route toward conformal electronics based on semiconducting van der Waals materials by combining high-throughput roll-to-roll mechanical exfoliation with commercially available temporary tattoo and waterslide decal transfer substrates. This approach enables the fabrication of ultrathin MoS₂-based electronic devices that can be transferred onto rough and curved surfaces such as skin, synthetic leather, and plant leaves. The resulting devices operate reliably after transfer and exhibit strong electronic and optoelectronic performance, including photodetectors with responsivities up to $\sim 3.5 \text{ A W}^{-1}$, thermistors with temperature coefficients of resistance of -2 to $-3.5\% \text{ }^\circ\text{C}^{-1}$, and ionic-gel-gated field-effect transistors with mobilities reaching $\sim 18 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$.

KEYWORDS: molybdenum disulfide (MoS₂), roll-to-roll exfoliation, conformal electronics, tattoo electronics, photodetectors, thermistors, field-effect transistors



The rapid development of flexible and conformal electronics has enabled the integration of electronic devices onto surfaces with complex topography, including human skin, textiles, and biological tissues.^{1–4} Such systems are central to emerging technologies in wearable health monitoring, human–machine interfaces, and biointegrated sensing, where devices must intimately adapt to soft, curved, and dynamically deforming surfaces while maintaining reliable electrical performance.

Two-dimensional (2D) van der Waals materials are particularly attractive for these applications because their atomic-scale thickness, mechanical flexibility, and absence of dangling bonds enable exceptional mechanical compliance and conformal contact with irregular surfaces.^{5–9} These properties allow 2D materials to preserve their electronic and optoelectronic functionality even under extreme bending or when interfaced with soft biological substrates. To date, most demonstrations of conformal bioelectronics based on 2D materials have relied on graphene, which has enabled highly sensitive epidermal sensors and bioelectronic interfaces.^{6,10–12} However, the absence of an intrinsic band gap in graphene limits its use in active electronic and optoelectronic components that require efficient switching or strong photoresponse. In contrast, semiconducting transition metal dichalcogenides (TMDs), such as MoS₂, possess intrinsic band gaps together with strong light–

matter interaction and good carrier mobility, making them attractive materials for active devices including transistors, photodetectors, and chemical sensors.^{13–17} Despite these advantages, the integration of semiconducting 2D materials into conformal and skin-interfaced electronic systems remains limited.

A major obstacle lies in the lack of scalable fabrication strategies that simultaneously provide high electronic quality, low cost, and large-area processability. Liquid-phase exfoliation enables scalable production of 2D materials and has been widely used to fabricate printed electronic devices, but the resulting films often exhibit limited electronic performance due to poor interflake connectivity and residual solvents trapped between nanosheets.^{18–20} At the other extreme, chemical vapor deposition (CVD) can produce high-quality continuous films with excellent electronic properties, yet it requires sophisticated

Received: March 9, 2026

Revised: June 3, 2026

Accepted: June 4, 2026

Published: June 11, 2026



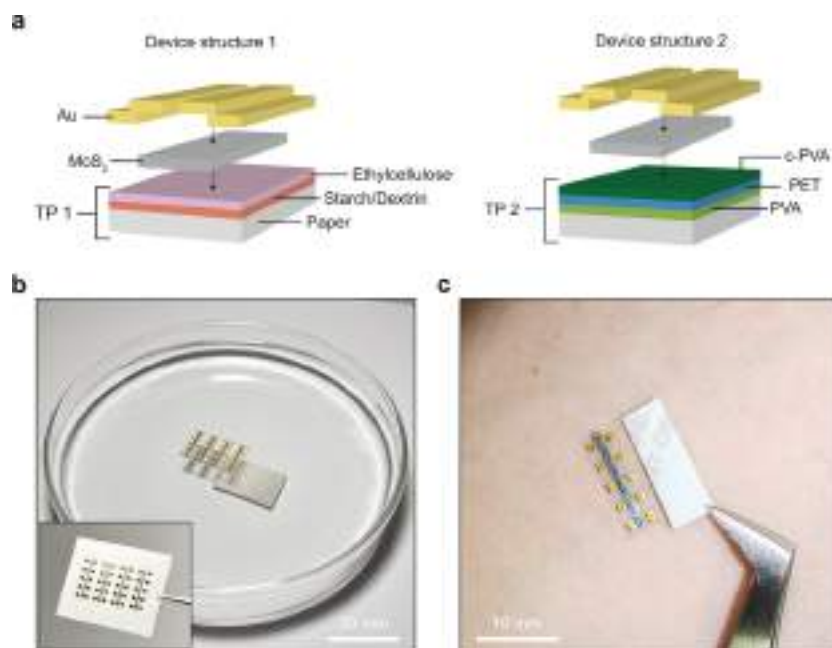


Figure 1. Delamination and transfer process for MoS₂ devices fabricated on tattoo and decal papers. (a) Cross-sectional schematic of the two different device structures composed of two different transfer papers: TheMagicTouch Tattoo 2.1 paper (left) and Hayes waterslide decal paper (right). In TheMagicTouch Tattoo 2.1 transfer paper, the device sits on a thin ethylcellulose film that can be released from the backing paper through a water-soluble starch/dextrin layer. In Hayes transfer paper, the device is most likely (the detailed proprietary composition is not disclosed by the manufacturer) supported by a partially cross-linked PVA layer (typically designed to increase ink adhesion) on a PET hydrophobic carrier substrate that can be released from the backing paper through a water-soluble PVA film. (b) Delamination of the fabricated MoS₂-based devices upon immersion in water using TheMagicTouch Tattoo 2.1 paper. The inset shows the as-fabricated devices on the paper before delamination. The floating device can then be scooped and transferred onto a substrate. (c) Transfer process for Hayes waterslide decal paper. The device is placed on the skin and moistened with a damp cloth, and the water-soaked backing paper is gently slid away to leave the decal adhered to the surface. More details about the transfer process are shown in Figure 2.

infrastructure and typically involves complex transfer processes that complicate integration onto unconventional substrates.^{21,22} Consequently, most demonstrations of conformal electronics based on semiconducting 2D materials remain limited to small-scale laboratory fabrication approaches.

Temporary tattoo transfer papers have recently emerged as attractive substrates for conformal electronics because they enable the transfer of ultrathin polymer-supported films onto curved and rough surfaces such as skin, glass, or plastics. These systems typically consist of a thin transferable polymer layer supported by a water-soluble sacrificial layer that releases upon wetting, allowing the film to conformally adhere to the target surface. Previous studies have successfully combined tattoo transfer substrates with organic semiconductors to fabricate epidermal electronic devices for physiological monitoring, as well as transferable components such as photodiodes, field-effect transistors, and solar cells.^{23–28} While these demonstrations highlight the versatility of tattoo-based transfer strategies for conformal electronics, they have largely relied on organic electronic materials, whose relatively poor charge transport properties limit device performance.

Here, we address these limitations by combining high-throughput roll-to-roll mechanical exfoliation of van der Waals materials with ultrathin temporary tattoo and waterslide decal transfer substrates to establish a scalable route toward conformal electronics based on semiconducting 2D materials. Using a recently reported roll-to-roll-like dry exfoliation strategy,²⁹ this approach produces large-area films composed of interconnected 2D semiconductor flakes with electronic properties superior to those typically obtained from solution-processed materials.

These films are integrated into ultrathin transferable platforms using commercially available decal-based substrates, enabling the fabrication of conformal photodetectors, thermistors, and ionic-gel-gated field-effect transistors that can be transferred directly onto rough and curved surfaces such as skin, synthetic leather, and plant leaves. The resulting devices exhibit high responsivity ($\sim 3.5 \text{ A W}^{-1}$), a large temperature coefficient of resistance (TCR) (from -2 to $-3.5\% \text{ }^\circ\text{C}^{-1}$) within the physiological range, and low-voltage transistor operation with mobilities reaching up to $\sim 18 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. By combining scalable production of semiconducting van der Waals materials with simple ultraconformal transfer strategies, this work extends tattoo-based electronics beyond organic systems and establishes a practical platform for high-performance wearable and biointerfaced devices.

RESULTS AND DISCUSSION

The fabrication of ultraconformal electronic devices was realized through the integration of mechanically exfoliated MoS₂ thin films with commercially available transfer papers. Specifically, two complementary transfer media were used: temporary tattoo transfer paper (TheMagicTouch 2.1 Tattoo) and waterslide decal paper (Hayes). Both papers are designed to release a thin hydrophobic polymer film upon wetting, enabling the transfer of printed patterns to curved and rough surfaces. Notably, in this work, the electronic devices are fabricated on the transferable film itself and then transferred onto the target surface, rather than transferring a printed pattern.

Tattoo paper and waterslide decal paper rely on a similar design that involves a hydrophilic backing paper providing

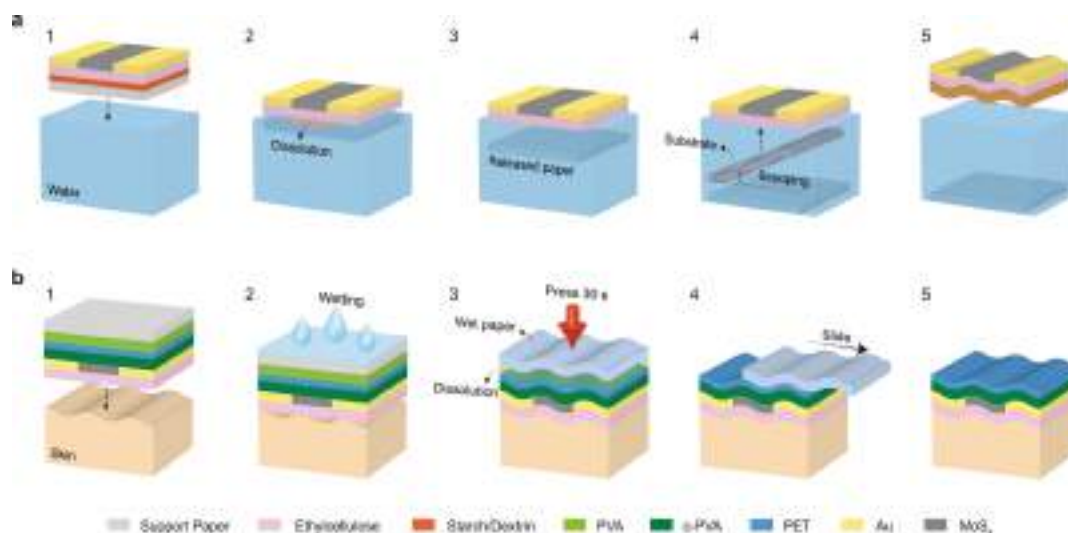


Figure 2. Step-by-step transfer methods for MoS₂-based tattoo devices using TheMagicTouch Tattoo 2.1 paper and Hayes waterslide decal paper. (a) The transfer process using TheMagicTouch Tattoo 2.1 paper involves immersing the paper in water (1) and allowing the fabricated device to delaminate and float (2–3). The floating device is then carefully scooped at a tilted angle and placed onto the target substrate (4–5). (b) The transfer process for Hayes waterslide decal paper starts with adhering the ethylcellulose-encapsulated device to the target substrate (1–2). After wetting, the backing paper is pressed for 30 s (3), allowing the PVA layer to partially dissolve (4), and then gently slid off, leaving the device on the target surface (5).

mechanical support, a water-soluble sacrificial layer enabling release, and a thin transferable hydrophobic polymer film carrying the printed or functional structures. Despite this similar architecture, the two transfer media differ in their composition. The structure and layer composition of tattoo paper and waterslide decal paper used in this study are represented in Figure 1a. From top to bottom, the tattoo paper (TheMagicTouch 2.1 Tattoo) consists of a thin ethylcellulose film, a water-soluble starch/dextrin layer, and a hydrophilic backing paper. Upon soaking, the intermediate layer dissolved and released the ethylcellulose film. Waterslide decal paper (Hayes) also consists of a multilayer structure, although the manufacturer does not disclose a detailed layer-by-layer description. However, previous patents on inkjet waterslide media suggest a generic architecture comprising a backing paper, a water-soluble poly(vinyl alcohol) (PVA)-based release layer, and a hydrophobic carrier film.^{30,31} According to our Raman measurements (Figure S1), the spectra of the released Hayes film show the characteristic peaks of PVA and polyethylene terephthalate (PET). We therefore infer that Hayes paper follows a similar design: a backing paper, a PVA water-soluble sacrificial layer, a PET hydrophobic carrier film, and an ink-receptive layer, most likely composed of cross-linked PVA.

The surface morphology and thickness of transferable films were analyzed via atomic force microscopy (AFM) (see Figure S2). For the ethylcellulose layer, we obtained an average thickness of 600 nm, which is consistent with the previously reported value.²⁷ In contrast, the transferred cross-linked PVA/PET film obtained from the Hayes paper is thicker and displays local thickness variations ranging from 1.8 to 2.2 μm . This inhomogeneity is inherent to the waterslide transfer process and is attributed to residual water-soluble PVA remaining on the transferred film. During transfer, the decal is applied face-down, such that the cross-linked PVA-based layer contacts the substrate while the water-soluble PVA layer remains on top after release. Although this upper PVA layer partially dissolves during soaking, it is not completely removed upon detachment of the backing paper, leading to spatial variations in the final film

thickness. A detailed description of the waterslide mechanism is provided below (see the discussion related to Figure 2).

The top polymer layer in both tattoo paper and waterslide decal paper served as a carrier platform for device fabrication, enabling subsequent transfer onto target surfaces. Device fabrication begins first with the deposition of MoS₂ thin films on the transfer media, which serve as a sensing (active) layer in our devices. These films were prepared via roll-to-roll mechanical exfoliation, a semiautomated process that enables continuous exfoliation of van der Waals crystals to produce large-area films from nanosheets.^{29,32} A well-connected percolating network of MoS₂ nanosheets was achieved by performing successive film transfers on substrates by a thermal release process. Further details of the exfoliation procedure and transfer process are provided in the “Materials and Methods” section. Optical microscope images obtained after each transfer step are shown in Figure S3, demonstrating the progressive formation of the flake network on each substrate during successive transfers. We performed a quantitative analysis of the evolution of the MoS₂ films on each transfer medium by extracting the coverage ratio after each transfer step. As shown in the coverage ratio plots in Figure S4, tattoo paper reaches near-complete surface coverage with fewer transfer steps than Hayes waterslide decal paper. Accordingly, 5–6 and 7–8 transfer steps were required to obtain films with long-range percolating flake networks on tattoo paper and waterslide decal paper, respectively.

AFM topography images of MoS₂ nanosheets after a single transfer onto tattoo and waterslide decal paper are presented in Figure S5a and b, respectively. The flake lengths were extracted from these images, and the corresponding distributions are shown in Figure S5c and d. Fitting these distributions to a log-normal function yields modal nanosheet lengths of 0.8 and 0.7 μm for tattoo paper and waterslide decal paper, respectively, with a mean value of 1.3 μm for both. These values are consistent with those previously reported for roll-to-roll exfoliated van der Waals crystals transferred onto rigid Si/SiO₂ substrates.^{29,32} The similar flake size distributions obtained on substrates with

different surface properties (e.g., roughness and elasticity) suggest that the dimensions of the nanosheets are determined mainly by the exfoliation dynamics rather than the substrate characteristics. Due to the surface roughness of the underlying substrates, AFM measurements did not allow reliable determination of individual flake thicknesses. Nevertheless, comparable thickness values to those previously obtained on SiO₂/Si substrates can still be expected, with reported modal thicknesses in the range of approximately 30–40 nm.^{29,32} Figure S6 shows Raman spectra acquired from MoS₂ flakes transferred onto tattoo and waterslide decal papers, which exhibit two prominent peaks corresponding to E_{2g}¹ and A_{1g} modes, located at 382 cm⁻¹ and 407 cm⁻¹, respectively.^{33,34} The absence of any noticeable peak shift or broadening indicates that the structural integrity of the MoS₂ flakes is well-preserved after transfer.

Transfer length method measurements were conducted on MoS₂ films with varying numbers of transfer cycles on tattoo paper to evaluate the effect of successive transfers on their electrical conductivity. For this purpose, films consisting of 2, 3, 4, and 5 transfer cycles with a width (*W*) of ~1 mm were prepared on tattoo paper, followed by the deposition of Au contact arrays with varying interelectrode spacings via thermal evaporation. Figure S7a gives resistance versus channel length plots for films with different numbers of transfer cycles. In each data set, the resistance decreases with decreasing channel length, while an overall reduction in resistance is observed with an increasing number of transfers, reflecting improved electrical conductivity arising from the progressive formation of percolating pathways across the film. The sheet resistance (*R_s*) for each data set was extracted from the slope of linear fits, where the slope corresponds to *R_s*/*W*. As shown in Figure S7b, the sheet resistance exhibits a decreasing trend with an increasing number of transfer cycles. After five transfers, the sheet resistance reaches ~10⁹ Ω □⁻¹, which is in good agreement with values reported for MoS₂ networks fabricated using other cost-effective techniques, such as liquid-phase exfoliation and abrasion-induced deposition.^{35–37}

The final device structures were obtained by transferring bar-shaped MoS₂ films onto the transfer papers, followed by the deposition of 80 nm thick Au contacts via thermal evaporation by using a commercial shadow mask (Ossila). The final configuration of the fabricated device arrays is shown in the inset of Figure 1b. For the transfer of the fabricated devices onto target surfaces, we followed two distinct transfer processes for the tattoo paper and waterslide decal paper. The transferable layer in tattoo paper was detached from the entire structure by soaking it in water (see Figure 1b) and then transferred onto the target surface by scooping. For waterslide transfer, the decal paper was placed face-down on the skin and wetted using a damp sponge or tissue. The water dissolves the underlying release layer, allowing the paper backing to slide off and leave the polymer film adhered to the skin (Figure 1c).

Figure 2a,b shows the step-by-step schematic illustrations of the transfer processes used for the integration of devices fabricated on the respective papers onto target surfaces. We further address the reader to Video S1 and S2 (Supporting Information) for demonstrations of the transfer procedures. In the case of tattoo paper (Figure 2a), the fabricated sample was placed in a Petri dish filled with deionized water and left to soak for approximately 1–2 min. During this period, the water-soluble sacrificial layer (starch/dextrin) gradually dissolves, allowing the backing paper to detach and sink. As a result, the freestanding, transferable ethylcellulose film carrying the device

starts to float on the water surface and becomes ready for transfer onto the target substrate. After the release, the desired target substrate was carefully immersed beneath the floating film using tweezers and aligned with respect to its position. The device film was then transferred onto the substrate by slowly lifting it through the water surface at a tilted angle of approximately 60°, allowing the film to smoothly adhere without trapping air bubbles or causing mechanical deformation. To further enhance conformal contact of the film with the substrate, a gentle flow of nitrogen gas was applied, ensuring adhesion and minimizing wrinkles or trapped water at the film/substrate interface. Finally, the sample was annealed at 70 °C for 10 min under ambient conditions to remove residual water and ensure full conformal contact with the transferred surface.

Prior to transferring devices from the water slide decal paper, the device-containing layer was encapsulated with an ethylcellulose layer obtained from tattoo paper (TheMagicTouch 2.1). This additional layer improves the adhesion of the transferred structure to the target substrate, particularly on rough surfaces. The encapsulation step can be carried out by using the same face-down transfer procedure described below. Alternatively, devices can be directly transferred without the ethylcellulose overlayer, still yielding fully functional devices, although with reduced adhesion.

The transfer of devices from waterslide decal paper was achieved by placing the transferable, ethylcellulose-covered device-containing side face-down onto the target substrate (see Figure 2b). After positioning, the backing paper was wetted with a small amount of deionized water (typically via a damp sponge or tissue) and gently pressed onto the target surface. After ~30 s, the sacrificial layer (PVA) partially dissolves, allowing the backing paper to be gently slid away, while the transfer layer remains securely adhered to the target surface. The same nitrogen-assisted drying and annealing procedure can also be applied to samples transferred onto substrates other than skin. It should be noted that, since the devices are sandwiched between the transferable layer and the target substrate after the transfer, direct electrical access to the electrodes is obstructed. Therefore, prior to the transfer, electrode regions should be pierced with a fine needle to obtain small vias. After transferring the device onto the target substrate, silver paste was applied to these openings to establish electrical contact between the device electrodes and the external contacts.

It should be noted that the scooping method used for tattoo paper is not applicable to waterslide decal paper, as complete self-delamination of the backing layer does not occur upon exposure to water. Conversely, applying the face-down transfer method to tattoo paper may induce mechanical damage, such as cracking of the gold electrodes due to the ultrathin nature of the ethylcellulose layer. Therefore, each transfer method is specifically suited to its corresponding transfer paper and enables a high transfer efficiency.

The two transfer approaches serve distinct and complementary roles, each offering specific advantages and limitations. The scooping transfer process used for devices fabricated on tattoo paper enables the gentle release of the device-containing ethylcellulose film with minimal mechanical stress, making it particularly suitable for fragile and ultrathin device structures. When the film is carefully lifted with the target substrate, wrinkle formation and air trapping at the film–substrate interface can be minimized. However, this method requires careful handling, as the floating film may drift on the water surface during the scooping step, requiring repeated repositioning to achieve

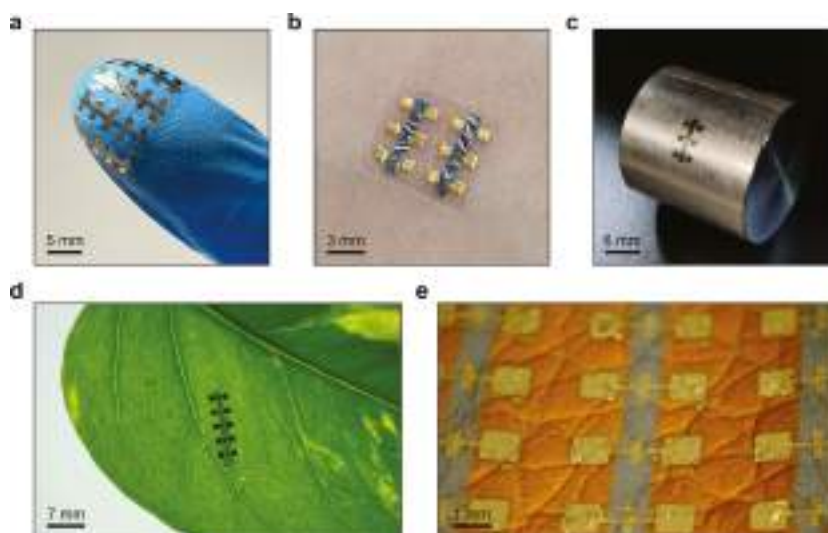


Figure 3. Conformal integration of MoS₂-based devices on diverse substrates. (a) Devices transferred onto a nitrile glove, illustrating adaptability to curved and textured surfaces. (b) Devices adhered to human skin, showing excellent conformability and adhesion. (c) Devices transferred onto a cylindrical metallic surface. (d) Devices successfully transferred onto a natural leaf, underscoring versatility for biointegrated applications. (e) Close-up of the devices on synthetic leather, demonstrating compatibility with artificial materials.

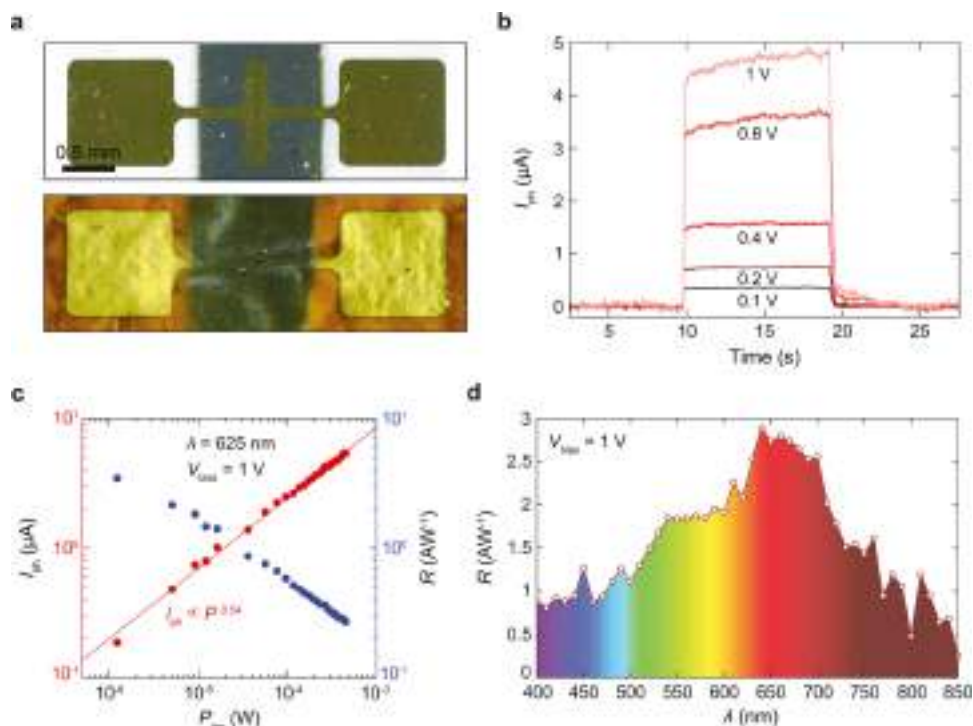


Figure 4. Optoelectronic performance of the MoS₂ photodetector fabricated on a waterslide decal paper after transfer onto synthetic leather. (a) Optical images in the top and bottom panels show a single device before transfer (on waterslide decal paper) and after transfer onto synthetic leather, respectively. (b) Time-resolved photocurrent measurements under different bias voltages (625 nm, 0.45 mW), illustrating stability and reproducibility. (c) Log–log plot of photocurrent (I_{ph}) versus light power (P_{inc}) and corresponding responsivities. Photocurrent shows a nonlinear dependence on illumination power with a power-law exponent of 0.54. (d) Spectral responsivity as a function of the light wavelength, highlighting the broadband sensitivity of the MoS₂ photodetector.

precise alignment. In contrast, the waterslide decal paper transfer method is simpler and more direct to implement. It enables straightforward positioning on the target substrate, making it well-suited for rapid and routine transfer processes, particularly in applications requiring simple handling and ease of use.

Accordingly, the selection of an appropriate transfer substrate can be critical for specific applications to achieve a reliable device integration and optimal performance. For instance, devices fabricated on tattoo paper are thinner than those fabricated on water slide decal paper, resulting in improved conformal adhesion and reduced risk of delamination. This makes them more suitable for applications requiring intimate mechanical

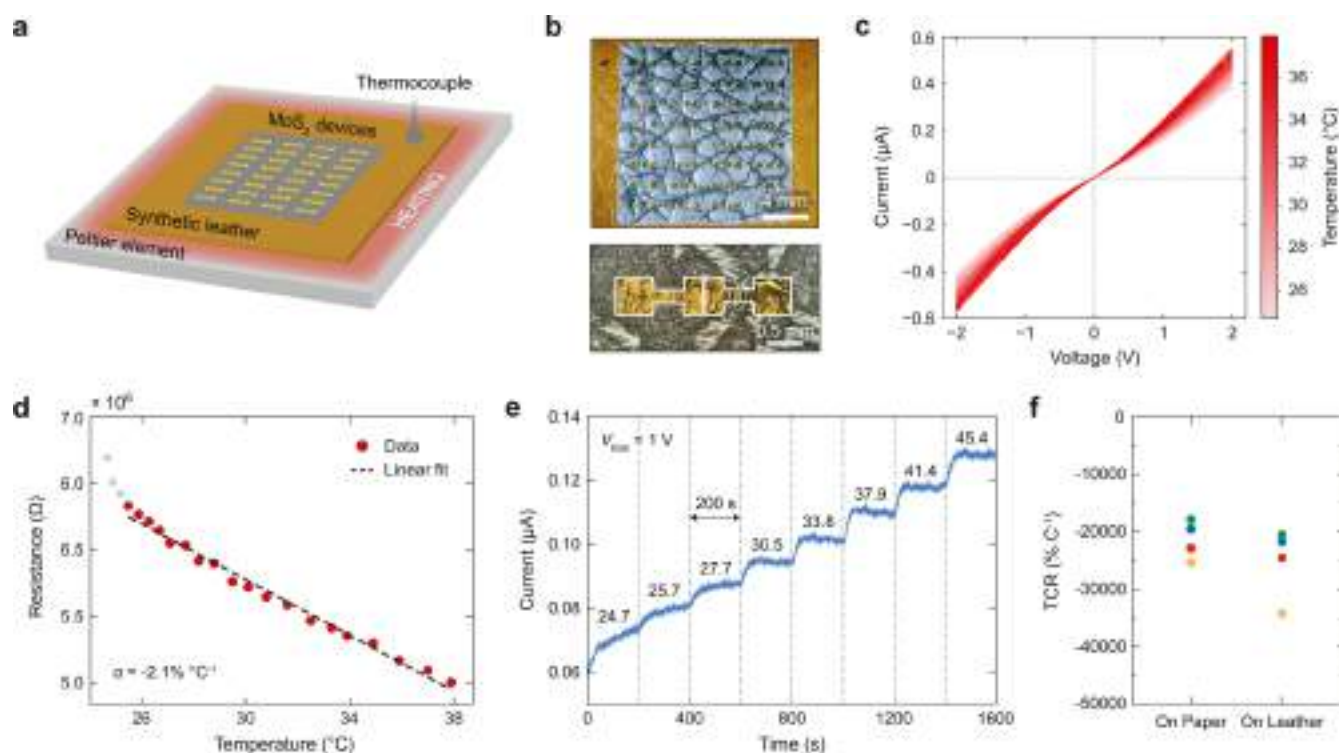


Figure 5. Temperature-dependent electrical characterization of MoS₂-based tattoo devices. (a) Simplified schematic of the experimental setup used for temperature-sensing measurements. (b) Optical images of the MoS₂ device array fabricated onto synthetic leather (top) and a magnified view of a single device (bottom). (c) Evolution of the *I*–*V* characteristics with increasing temperature, from room temperature up to 38 °C. (d) Temperature dependence of electrical resistance during heating. A high TCR value of $-2.1\% \text{ } ^\circ\text{C}^{-1}$ was extracted from the linear fit (black dashed line). Gray dots correspond to the data excluded in the linear fit. (e) Real-time monitoring of the current response during stepwise heating. The numbered points within the plot correspond to the temperature values recorded once the system reached thermal equilibrium at each step. (f) Comparison of TCR values from four devices, obtained before transfer (on tattoo paper) and after transfer (on synthetic leather).

coupling, such as skin-mounted or flexible surface-integrated devices. In contrast, devices transferred using waterslide decal paper without an ethylcellulose overlayer allow direct contact between the device channel and the target surface. This feature is particularly advantageous for wearable biosensing applications as it enables real-time monitoring of biochemical markers.

Next, to demonstrate adaptability and conformal integration capability, we transferred the tattoo paper and waterslide decal paper-based devices onto a range of unconventional substrates with different surface textures, curvatures, and mechanical properties (see Figure 3). Figure 3a shows the full device array integration onto a nitrile glove, indicating the seamless conformity of the transferred film onto the soft, elastic, and high-friction surface. Figure 3b shows the physical adhesion of the film onto human skin, a biologically relevant surface known for its rough texture, curvature, and moisture content. The film maintained intimate contact with the skin without any visible delamination or cracking even under repeated stretching (see Video S3). Figure 3c shows the transfer on the curved metallic substrate, emphasizing the process's compatibility also with rigid, curved surfaces. The zoomed-in optical microscope image in Figure S8 shows that the device conforms well to the surface geometry, with no wrinkling, air gaps, or delamination. Figure 3d shows the integration of the devices onto a natural leaf, a fragile, moisture-containing, and irregular surface. Figure 3e highlights the successful adhesion of the device array onto synthetic polyurethane (PU) leather with a lychee-pattern surface that reproduces the grain and compliance of natural leather. Because of its pronounced surface roughness, soft mechanics, and

complex topography (often exceeding those of human skin), we employed this PU leather in the following sections as an artificial testing platform for device characterization. This surrogate surface effectively captures key mechanical and topographical features of skin while providing a safe, reproducible, and standardized environment for evaluating transfer quality and device performance. The scanning electron microscopy (SEM) image presented in Figure S9 demonstrates the conformal adhesion of a tattoo device on the synthetic leather.

Although the adaptability of the devices to various platforms is illustrated using a single transfer medium in Figure 3, it should be emphasized that we do not observe any fundamental limitation for transferring the devices onto the demonstrated substrates using either transfer papers. Both transfer approaches are broadly applicable across these platforms and enable a low-cost, high-throughput transfer of conformable devices.

In the following sections, we systematically examine the performance of the devices after transferring them onto various platforms, such as synthetic leather and leaves, to evaluate their ability to operate reliably on substrates with different surface properties. A series of measurements were performed to characterize their sensitivity to light and temperature fluctuations and to evaluate their performance when they are operated as field-effect transistors.

Figure 4 reveals the detailed optoelectronic performance of a waterslide decal paper-based device after its transfer onto synthetic leather. Optical images of a single device on waterslide decal paper and after transfer onto synthetic leather are shown in Figure 4a. Figure 4b presents the time-resolved photocurrent

response measured under different bias voltages (V_{bias}), under illumination at a wavelength of 625 nm and a power of 0.45 mW. The device exhibits stable and reproducible switching between illuminated and dark states, indicating that the transferred device retains its photoswitching capability. The response time of the device was estimated as <40 ms, limited by the response time of the read-out electronics. Figure 4c shows the dependence of photocurrent (I_{ph}) on the incident light power, plotted on a log–log scale. The photocurrent exhibits a clear power-law dependence, following $I_{\text{ph}} \propto P^\alpha$, with a power exponent $\alpha = 0.54$.³⁸ This sublinear behavior is typical for photodetectors based on 2D materials, where processes such as trap state filling or recombination through defect states influence the carrier dynamics.³⁹ The corresponding responsivity (R) values can be calculated using the formula:

$$R = \frac{I_{\text{ph}} A_{\text{spot}}}{P_{\text{inc}} A_{\text{eff}}}$$

where I_{ph} is the measured photocurrent, P_{inc} is the total incident light power, A_{spot} is the total area of the illumination spot, and A_{eff} is the effective device area, which is obtained by multiplying the total area of the device photoactive channel with a factor c . Here, c is a correction factor ($0 < c \leq 1$) representing the fraction of the channel covered by MoS₂ flakes. This parameter was extracted from optical coverage analysis. The decreasing incident light intensity causes a gradual increase in the responsivity, which reaches a maximum value of 3.5 A W⁻¹ at the lowest illumination power of $\sim 1 \mu\text{W}$. The spectral responsivity of the device, shown in Figure 4d, confirms that the photodetector responds broadly across the visible spectrum (expected for multilayer MoS₂), with a peak responsivity in the 650–700 nm range, which is consistent with the direct band-to-band transition of MoS₂. To demonstrate the reproducibility and robustness of the method, several additional photodetector devices were transferred onto both synthetic leather and a curved metallic surface. Accordingly, 9 out of 10 devices remained functional and exhibited consistent photoresponse behavior after their transfer (see Figure S10). Furthermore, devices fabricated on two different transfer papers were also transferred onto leaves and operated as photodetectors, with the corresponding results presented in Figures S11–S14.

Next, the devices were operated as thermistors, where their resistance variation with the temperature was monitored to evaluate the thermal sensitivity. For thermistor applications, tattoo paper was preferred owing to its better conformability compared to waterslide decal paper. Improved conformal contact between the device and the target surface enhances heat transfer, enabling faster thermal equilibration and more-accurate temperature sensing. After transferring MoS₂ films onto tattoo paper, Au electrodes were deposited through a custom-designed shadow mask via thermal evaporation.⁴⁰ This custom-made mask enables the fabrication of 32 individual devices within a compact 1.5 cm² area, allowing high-throughput characterization under identical fabrication and measurement conditions. The completed films were subsequently transferred onto synthetic leather by using the scooping technique.

For temperature-dependent electrical characterization, the MoS₂-based tattoo devices were placed on a Peltier element, which was used as a controllable heating platform. The experimental setup is schematically illustrated in Figure 5a, and real images of the prepared sample are given in Figure 5b. The Peltier element was operated in reverse-polarity config-

uration by a DC power supply (TENMA 72-2715) to induce heating. The applied voltage to the Peltier element was gradually increased in discrete steps to modulate the substrate temperature. At each temperature point, current–voltage (IV) measurements were performed to monitor the device response to the increasing temperature. Note that, after each temperature increment, a 30 s stabilization period was allowed to ensure thermal equilibrium before recording the IV characteristics. A thermocouple was positioned in close proximity to the device region on top of the synthetic leather surface to accurately record the local temperature during measurements.

Figure 5c clearly illustrates the impact of temperature on the I – V characteristics, showing that higher temperatures lead to increased current levels, which are more likely associated with a combination of thermally activated charge carriers, enhanced carrier injection from the contacts, and thermally activated hopping between interconnected MoS₂ nanosheets, originating from reduced junction resistance between adjacent flakes. Due to the supralinear I – V characteristics, which are primarily attributed to bias-induced Joule self-heating of the ultrathin MoS₂ tattoo thermistors, the resistance values were extracted by linear fitting within a small voltage range where the curves are approximately linear. Figure 5d presents the resistance variation as the temperature was swept in the range of 25 to 38 °C, which is typically the range used for body-temperature sensors and wearable thermistors. The device resistance exhibited a linearly decreasing trend with increasing temperature, indicating negative temperature coefficient (NTC) thermistor behavior of semiconducting MoS₂.⁴¹

Temperature coefficient of resistance (TCR), sometimes referred to as sensitivity, is a key parameter to quantitatively describe the sensitivity of the device's resistance to temperature and can be expressed as

$$\text{TCR} = \frac{1}{R_{\text{ref}}} \frac{dR}{dT}$$

where R_{ref} is the reference resistance, measured at a temperature closest to 36.5 °C, which is within the typical range of normal human body temperature. By performing a linear fit, the TCR value was determined to be $-2.1\% \text{ } ^\circ\text{C}^{-1}$. This TCR value is higher than previously reported MoS₂-based temperature sensors as given in Table S1, which also summarizes previously reported TCR values for temperature sensors based on different sensing materials implemented on flexible and stretchable platforms. Figure 5e presents the real-time current response of the MoS₂ tattoo thermistor during consecutive heating, with the temperature swept between 24.7 and 45.4 °C. After each temperature increment, a 200 s holding period was applied to perform the read-out before jogging to the next temperature value. The current exhibited a rapid, stepwise increase with rising temperature, maintaining a stable level at each temperature plateau. Figure 5f summarizes the TCR values extracted from four MoS₂-based tattoo devices, both before transfer (on tattoo paper) and after transfer onto synthetic leather. Across both substrates, the devices demonstrated TCR values of comparable magnitude, indicating uniform sensing characteristics. The temperature-dependent resistance of additional devices following transfer onto synthetic leather is shown in Figure S15.

We further explore the use of the MoS₂ conformal devices as proof-of-concept FETs. To build top-gated MoS₂-based tattoo FETs, we used a commercially available ionic conductive hydrogel (see Materials and Methods), which was placed on top

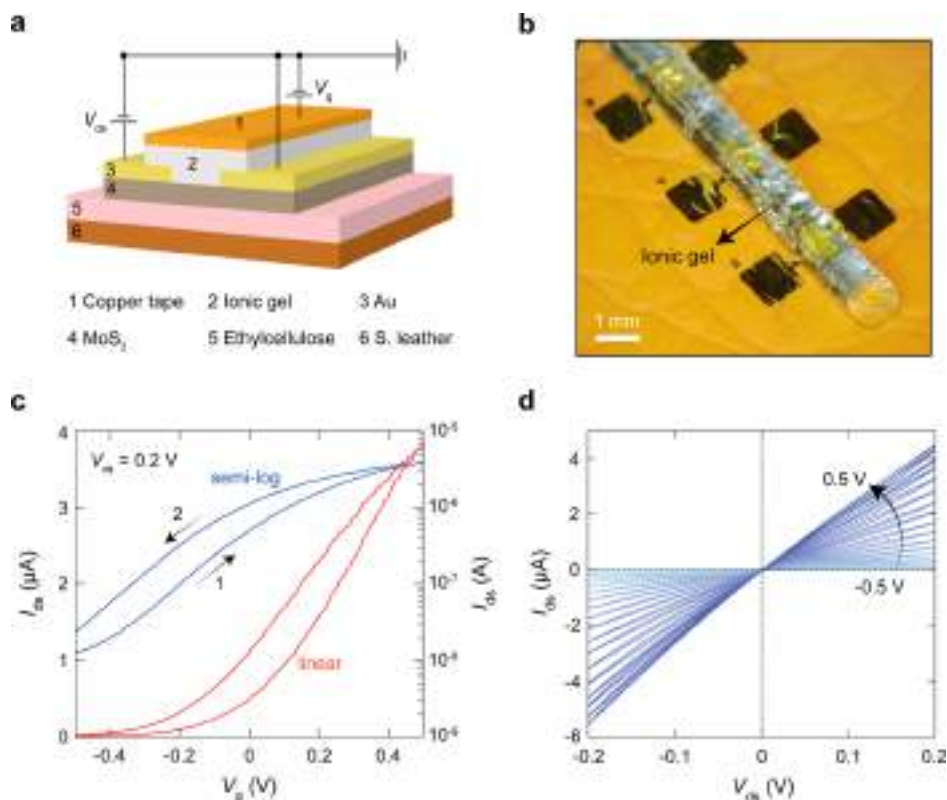


Figure 6. Ionic gel-gated MoS₂ tattoo-FET on synthetic leather. (a) 3D cross-sectional schematic illustrating the layer-by-layer FET structure. (b) Optical images of an ionic-gel-transferred tattoo device on synthetic leather. (c) Semilog and linear-scale plot of transfer characteristics under forward (1) and backward (2) gate voltage sweep for a V_{ds} of 0.2 V. (d) Output characteristic curves under gate voltage sweeps from -0.5 to 0.5 V.

of the tattoo paper devices transferred onto synthetic leather to serve as an electrolyte–gate dielectric. The ionic gel, a composite of an ionic liquid and a flexible polymer matrix, exhibits electrochemical behavior similar to that of the ionic liquid while offering greater mechanical robustness and solid-like characteristics. When a gate voltage is applied, the mobile positive and negative ions within the gel migrate toward the gate electrode and MoS₂ interfaces, respectively, forming two electrical double layers, one at each interface. The electrical double layer at the gel/semiconductor interface acts as an ultrathin gate dielectric, creating a large interfacial capacitance and enabling efficient tunability in channel conductance within a small gate voltage range.

It is important to note that present devices are a proof-of-concept demonstration; we did not specifically engineer the ionic gate, and the current implementation employs a relatively thick ionic gel. Nevertheless, we are confident that the device architecture can be significantly optimized in future studies by research groups with expertise in ionic gating materials.

Tattoo FETs were obtained by attaching a thin, strip-shaped cut piece of ionic gel onto a single-line device array transferred onto synthetic leather, selectively covering the channel regions to ensure proper gating while minimizing contact with the source and drain electrodes. Later, a copper tape was placed on top of the gel to serve as the gate electrode. Figure 6a,b shows the device composition and an optical image of the devices after ionic gel transfer (prior to attaching the copper tape gate electrode on top), respectively. Figure 6c shows the transfer characteristics for one of the fabricated tattoo FETs in semilog and linear scale, measured at a drain–source voltage (V_{ds}) of 0.2 V. Due to the slow ion migration within the ionic gel under the

applied gate voltage, the gate voltage was swept in 0.01 V steps every 3 s. The device exhibits typical n-type transistor behavior and can be operated within a small gate voltage range, with a threshold voltage (V_{th}) of ~ 10 mV, which was extracted by linear extrapolation of the linear-scale $I_{ds} - V_g$ curve to the V_g axis. The device shows an on/off ratio (I_{on}/I_{off}) of 3×10^2 , with a subthreshold slope of 254 mV dec^{-1} . The effective field-effect mobility (μ) can be estimated from the equation

$$\mu = g_m \frac{L}{W_{\text{eff}} C_G V_{ds}}$$

where $g_m = (\partial I_{ds} / \partial V_g)$ is the transconductance, V_{ds} is the drain–source voltage, C_G is the gate capacitance of the ionic gel per unit area, and L and W_{eff} are the channel length and effective channel width, respectively. W_{eff} was determined by multiplying the total channel width, W , by a flake coverage factor of c , which is 0.8 for the corresponding device. The capacitance of the ionic gel was measured as $9.6 \times 10^{-3} \text{ F/m}^2$ using RC circuit measurements, which will be reported in a forthcoming study. This measured areal capacitance is consistent with values commonly reported for ionic gels and ionic liquids.^{42–45} μ was estimated using the maximum transconductance value extracted from the transfer curve and was obtained to be $1.2 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for this device. Figure 6d shows the gate-dependent IV curves of the tattoo FET, where the gate voltage was swept from -0.5 to 0.5 V. The more linear, ohmic behavior observed under ionic gel gating is likely associated with electrostatic carrier accumulation induced by the ionic gel near the MoS₂/electrode interface. The increased local carrier density can enhance carrier injection from the contacts and reduce the effective contact resistance. In addition, ionic-gel-induced electrostatic doping of the MoS₂ channel may

further contribute to the enhanced device conductivity. It should also be noted that the FET measurements were performed within a significantly smaller V_{ds} compared to the photodetector and thermistor measurements in order to minimize ionic leakage currents and avoid undesired electrochemical reactions within the ionic gel. This reduced operating bias additionally suppresses self-heating effects and contributes to the more linear current–voltage characteristics observed in the FET devices.

To assess the device-to-device variability and reproducibility of the method, we have characterized 6 FETs in total under ionic gel gating on synthetic leather. Key performance parameters for all devices are summarized in Table S2, demonstrating that their overall electrical characteristics are closely aligned with only modest variation. The extracted mobilities from the transfer curves (see Figure S16) span from a minimum of $0.23 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ to a maximum of $17.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, yielding an average mobility of $4.55 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. Narrow gate window operation yields extremely small V_{th} values with minimal variation, ranging from -41 mV to 10 mV (see Figure S17), which is important for the development of flexible and wearable platforms that operate with low energy consumption. In our previous study on roll-to-roll mechanically exfoliated MoS_2 films, the highest measured mobility was $1.36 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. However, those devices employed a conventional back-gate SiO_2 configuration, where the relatively low gate capacitance limits electrostatic modulation of the channel conductivity. In contrast, the significantly higher capacitance of ionic gel dielectrics enables much stronger gate coupling and more efficient carrier modulation, resulting in substantially enhanced mobility values. It should be noted that the mobility values reported here for the ionic gel-gated devices are among the highest reported for MoS_2 nanosheet-network-based FETs.^{42,46–50}

As an alternative to ionic gel gating, ethylcellulose layers can serve as a gate dielectric to modulate the channel conductivity of devices on tattoo paper.²⁸ However, the modulation is weaker, likely due to the lower capacitance of the ethylcellulose layer ($5.3 \times 10^{-5} \text{ F/m}^2$) and the increased dielectric thickness arising from the double-layer ethylcellulose structure present in this device configuration needed to avoid gate leakage. The reader is referred to Figure S18 for further details on the device architecture and FET characterization.

Even when a uniform areal coverage is achieved through iterative film transfer, devices based on flake-network films can still exhibit significant variation in their electronic properties. This variation is primarily associated with the percolative nature of charge transport through the interflake junction network. In films produced by methods such as roll-to-roll mechanical exfoliation, electrochemical exfoliation, and liquid-phase exfoliation, current flows through percolation pathways defined by flake–flake contacts. The resistance of these pathways is largely governed by factors such as the overlap area between flakes, their relative alignment, and the lateral size of the flakes. When percolation pathways consist of well-aligned flakes with large overlap areas, or when they are formed by larger flakes that reduce the number of junctions along the conduction path, the overall device resistance is lowered. However, because these films are composed of randomly distributed flakes (each transfer introduces additional randomness into the network morphology), the resulting percolation network can vary significantly from device to device. Consequently, some devices may be dominated by highly resistive junctions, while others benefit from more conductive pathways. This intrinsic variability in the

percolation network is a key factor leading to fluctuations in the measured mobility and overall electronic performance.

The device performance could be further improved by achieving better control over nanosheet size distribution by optimizing exfoliation parameters (e.g., pressure and speed) or improving deposition processes to obtain better network uniformity.^{35,42} In combination with postprocessing techniques such as thermal annealing, compression/calendering, or covalent functionalization of the nanosheet network, device-to-device variations could be further reduced through improved interflake coupling and an enhanced degree of nanosheet alignment, leading to more uniform percolative charge transport across the large-area films.^{18,51–53}

Figure 7 presents reported μ values for transistors fabricated on flexible or stretchable platforms as a function of V_{th} , including

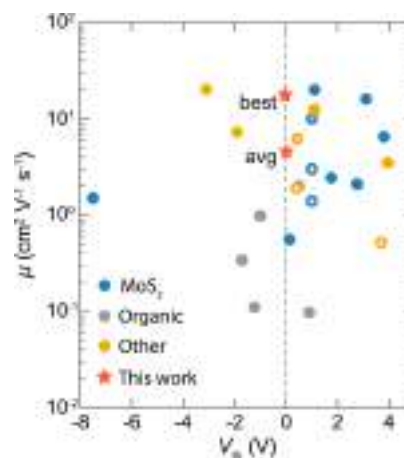


Figure 7. Benchmarking across flexible and stretchable FETs. The plot shows the mobility (μ) as a function of threshold voltage (V_{th}), comparing the average and highest values from our devices with those reported for transistors fabricated on flexible or stretchable substrates. For clarity, the devices are classified according to their active channel materials: MoS_2 (blue), organic polymers (gray), and other inorganic and carbon-based materials (orange). Open symbols denote ionic gated devices. Data points marked with an asterisk (*) indicate studies in which the V_{th} was not reported and was instead extracted by us from the transfer characteristics provided in the original publications.

our average and best values for direct comparison. The devices were classified into groups such as MoS_2 , organic polymers, and other inorganic and carbon-based materials, to separate materials exhibiting fundamentally different charge-transport mechanisms. Additionally, ionic gated transistors are represented with open symbols to distinguish them from devices gated with solid-state dielectrics, which operate based on fundamentally different principles. It is worth noting that our values were achieved without any postfabrication treatments to improve device performance, such as acid treatments or high-temperature annealing, and without performing measurements under vacuum or encapsulated conditions. Accordingly, our μ values are significantly higher than those reported for organic field-effect transistors and also exceed those of ionic gated networks. The average μ value we obtained is on the same order of magnitude as the highest values reported in the literature.^{54–56} Moreover, the V_{th} values of our devices are among the lowest reported for comparable flexible and stretchable FET platforms. A more comprehensive summary

of the reported FET parameters for each study included in Figure 7 is provided in Table S3.

CONCLUSIONS

In this work, we demonstrate that combining high-throughput roll-to-roll mechanical exfoliation of MoS₂ with commercially available temporary tattoo and waterslide decal transfer papers provides a low-cost and scalable route for fabricating conformal devices. Unlike liquid-phase exfoliation followed by printing (where residual solvents, poor interflake connectivity, and limited film uniformity often compromise electronic performance) and unlike CVD-grown films (which require expensive infrastructure and complex transfer processes for the as-grown films), roll-to-roll mechanical exfoliation provides dry, interconnected MoS₂ films that can be readily integrated into ultrathin, transferable platforms. By leveraging this material platform together with simple decal-based transfer strategies, we fabricate ultraconformal devices that operate reliably on rough and compliant surfaces. The resulting tattoo-based photo-detectors, thermistors, and ionic gel-gated transistors exhibit high responsivity, large temperature coefficients of resistance, and low-voltage, high-mobility transistor operation.

METHODS/EXPERIMENTAL SECTION

Device Fabrication

MoS₂-based tattoo devices were fabricated on temporary tattoo papers and waterslide decal papers, which were supplied from TheMagic-Touch Spain S.L. (www.themagictouch.es) and Hayes Paper Co. (www.hayespaper.com), respectively. The MoS₂ thin films were obtained through the roll-to-roll mechanical exfoliation of a natural molybdenite mineral (Molly Hill Mine, Quebec, Canada). This system is based on continuous exfoliation of bulk van der Waals crystals through two rolling cylinders in contact, wrapped with adhesive Nitto tape, enabling the peeling of crystal layers as the cylinders rotate. The roll-to-roll exfoliation process was carried out for 1 min.

After exfoliation, bar-shaped MoS₂ films were patterned on the transfer papers by successively transferring flakes from Nitto tape onto tattoo paper or waterslide decal paper using a bar-shaped stencil mask. The stencil mask was fabricated from a 100 μm thick Mylar sheet using a smart cutting machine (Cricut Maker 3). The dimensions and spacing of the bar-shaped openings were designed to match the geometry and column-to-column spacing of the 4 × 5 electrode array in the commercial shadow mask (Ossila) used for electrode deposition, thereby enabling accurate alignment of the electrode channels with the films during mask placement.

After attaching the flake-containing Nitto tape to the tattoo paper (waterslide decal paper), the samples were annealed on a hot plate at 70 °C (100 °C) for 5 min to induce thermal release of the flakes. The narrow openings of the stencil mask can hinder proper contact between the tape and the substrate. To ensure complete attachment, gentle pressure was applied over the openings using a cotton swab shortly after placing the sample on the hot plate, once the substrate had begun to warm.

After annealing, the Nitto tape was removed immediately upon removing the sample from the hot plate, while still warm. This is particularly important to prevent tearing of the transferable polymer films on the paper. The transfer process was repeated to form a continuous, interconnected film of overlapping flakes. Finally, source–drain electrodes were obtained by depositing 80 nm thick Au contacts via thermal evaporation using commercial (Ossila) or home-built shadow masks.

For FET characterization, a commercially available conductive hydrogel (C100AE EMS hydrogel pads, model JP-EMS-11) of the type used in TENS (transcutaneous electrical nerve stimulation) pads was transferred onto the semiconducting MoS₂ channel to form an ionic interface layer. These hydrogels consist of water-rich polymer networks

(typically PVA-based water gels) containing mobile ions (dissolved salts), providing stable ionic conductivity and conformal contact with soft surfaces. To obtain the gate contact, copper tape was attached on top of the ionic gel.

Characterization

The atomic force microscopy images were taken in dynamic mode with a resonance frequency of 76 kHz using a cantilever oscillation amplitude of 1.7 V. The tip used was a PPP-FMR-50 from nanosensors, and the analysis was done with a commercial AFM from Nanotec.

Field-emission scanning electron microscopy images were collected on an FEI Nova NanoSEM 230 scanning electron microscope with a Schottky field-emission gun equipped with a W-ion source, using aluminum as a support.

For Raman measurements, transferable hydrophobic polymer films from tattoo paper and waterslide decal paper were deposited onto SiO₂/Si substrates and annealed at 80 °C for 15 min. The Raman spectra of the transferred films were then acquired under ambient conditions using a confocal Raman microscope (MonoVista CRS+, Spectroscopy & Imaging GmbH) with 532 nm excitation from a continuous wave (CW) solid-state laser. A 300 lines/mm diffraction grating was used, providing a spectral resolution of 6 cm⁻¹. The incident light power was set to 0.54 mW and focused through a 50× objective (NA = 0.75), providing a spot size of 2 μm.

Electrical Measurements

All electrical measurements were performed under atmospheric pressure at room temperature using a home-built probe station. In the FET measurements, the source–drain terminals were biased using probes connected to a Keithley 2450 source-meter unit, while a controlled gate-voltage sweep was applied to the gate terminal through a separate probe using two programmable benchtop power supplies (Tenma 72-2715) connected in back-to-back configuration.

Optoelectronic Measurements

Photocurrent measurements under varying power and bias conditions were performed using a fiber-coupled LED source at a wavelength of 625 nm (Thorlabs M625F2), operated by a LED driver (Thorlabs LEDD1B).⁵⁷ The LED output power was varied by tuning the LED drive current with a Tenma power supply unit (model 72-2715). Wavelength-dependent photocurrent measurements were conducted using a tunable xenon lamp source (Bentham TLS120Xe), sweeping the incident light across the visible range from 400 to 850 nm in 10 nm increments.

ASSOCIATED CONTENT

Data Availability Statement

The data sets generated and/or analyzed during the current study will be made publicly available in the Zenodo repository through our community page (2D Foundry community at Zenodo) at: <https://zenodo.org/communities/2dfoundry>. All relevant data supporting the findings of this study will be accessible upon publication.

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsnano.6c04448>.

Figure S1: Raman spectra of transferable polymer films of tattoo and waterslide decal papers; Figure S2: AFM analysis of the surface morphology and thickness of ethylcellulose and cross-linked PVA + PET films; Figure S3: evolution of the MoS₂ film with successive transfers on tattoo paper (TheMagicTouch Tattoo 2.1) and waterslide decal paper (Hayes); Figure S4: substrate coverage as a function of the number of transfer steps; Figure S5: AFM characterization of MoS₂ flakes on tattoo paper and waterslide decal paper; Figure S6: Raman characterization of transferred MoS₂ flakes; Figure S7: TLM analysis of

MoS₂ films on tattoo paper; Figure S8: zoomed-in optical microscope image of the MoS₂-based tattoo device on a curved metallic surface; Figure S9: SEM image of a tattoo device on synthetic leather; Figure S10: photoresponse characteristics of photodetectors on a curved metallic surface and synthetic leather; Figure S11: electrical and photoresponse characteristics of a waterslide decal paper-based photodetector before and after transfer onto a leaf; Figure S12: power-dependent photocurrent measurements for waterslide decal paper-based photodetectors on a leaf; Figure S13: electrical and photoresponse characteristics of a tattoo paper-based photodetector before and after transfer onto a leaf; Figure S14: power-dependent photocurrent measurements for tattoo paper-based photodetectors on a leaf; Figure S15: temperature-dependent resistance of devices transferred onto synthetic leather; Table S1: literature summary of reported temperature sensors; Table S2: key device performance metrics of ionic gel-gated MoS₂ tattoo FETs on synthetic leather; Figure S16: transfer characteristics of ionic gel-gated MoS₂ tattoo FETs; Figure S17: linear-scale transfer characteristics of ionic gel-gated MoS₂ tattoo FETs; Figure S18: gating tattoo devices through ethylcellulose; and Table S3: literature summary for FETs demonstrated on flexible and stretchable platforms (PDF)

Video S1: scooping transfer of a MoS₂ device fabricated on TheMagicTouch Tattoo 2.1 transfer paper (MP4)

Video S2. Waterslide transfer of a MoS₂ device fabricated on Hayes waterslide decal paper (MP4).

Video S3: stretching test of a MoS₂ device transferred onto human skin (MP4)

AUTHOR INFORMATION

Corresponding Authors

Yigit Sozen – 2D Foundry Research Group, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid E-28049, Spain; Email: yigit.sozen@csic.es

Andres Castellanos-Gomez – 2D Foundry Research Group, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid E-28049, Spain; orcid.org/0000-0002-3384-3405; Email: andres.castellanos@csic.es

Authors

Esteban Zamora-Amo – 2D Foundry Research Group, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid E-28049, Spain

Juan J. Riquelme – 2D Foundry Research Group, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid E-28049, Spain

Complete contact information is available at: <https://pubs.acs.org/10.1021/acsnano.6c04448>

Author Contributions

Y.S. lead data curation; formal analysis; investigation; methodology; the original draft writing. E.Z. supported data curation, formal analysis, investigation, and methodology. J.J.R. supported resources, supervision and methodology. A.C.-G. lead conceptualization, funding acquisition, project administration, resources, supervision, supported the methodology, and original draft writing. All authors contributed equally to review and editing.

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

The authors thank Dr. Carmen Munuera (ICMM-CSIC) for her support with the AFM measurements and for useful discussions throughout this work. Grant PRE2021-098348 funded by MICIU/AEI/10.13039/501100011033 and by ESF+, and Grant PID2023-151946OB-I00 funded by MICIU/AEI/10.13039/501100011033 and by ERDF/EU. This work also received support from the European Commission—NextGenerationEU (Regulation EU 2020/2094) through CSIC's Quantum Technologies Platform (QTEP). The authors acknowledge funding from the European Research Council (ERC) through the ERC-2024 SyG SKIN2DTRONICS project (Grant Agreement No. 101167218). J.J.R. was hired under the Generation D initiative, promoted by Red.es, an entity attached to the Ministry for Digital Transformation and the Civil Service, aimed at attracting and retaining talent through grants and training contracts and financed by the Recovery, Transformation and Resilience Plan through the European Union NextGenerationEU funds. ICMM-CSIC authors also acknowledge support from the Severo Ochoa Centres of Excellence program through Grant CEX2024-001445 S, funded by MICIU/AEI/10.13039/501100011033.

REFERENCES

- (1) Kim, D.-H.; Lu, N.; Ma, R.; Kim, Y.-S.; Kim, R.-H.; Wang, S.; Wu, J.; Won, S. M.; Tao, H.; Islam, A.; Yu, K. J.; Kim, T.; Chowdhury, R.; Ying, M.; Xu, L.; Li, M.; Chung, H.-J.; Keum, H.; McCormick, M.; Liu, P.; Zhang, Y.-W.; Omenetto, F. G.; Huang, Y.; Coleman, T.; Rogers, J. A. Epidermal Electronics. *Science* **2011**, *333* (6044), 838–843.
- (2) Rogers, J. A.; Someya, T.; Huang, Y. Materials and Mechanics for Stretchable Electronics. *Science* **2010**, *327* (5973), 1603–1607.
- (3) Liu, S.; Rao, Y.; Jang, H.; Tan, P.; Lu, N. Strategies for Body-Conformable Electronics. *Matter* **2022**, *5* (4), 1104–1136.
- (4) He, T.; Wang, J.; Hu, D.; Yang, Y.; Chae, E.; Lee, C. Epidermal Electronic-Tattoo for Plant Immune Response Monitoring. *Nat. Commun.* **2025**, *16* (1), 3244.
- (5) Kireev, D.; Kutagulla, S.; Hong, J.; Wilson, M. N.; Ramezani, M.; Kuzum, D.; Ahn, J.-H.; Akinwande, D. Atomically Thin Bioelectronics. *Nat. Rev. Mater.* **2024**, *9* (12), 906–922.
- (6) Choi, C.; Lee, Y.; Cho, K. W.; Koo, J. H.; Kim, D.-H. Wearable and Implantable Soft Bioelectronics Using Two-Dimensional Materials. *Acc. Chem. Res.* **2019**, *52* (1), 73–81.
- (7) Kim, J.; Lee, Y.; Kang, M.; Hu, L.; Zhao, S.; Ahn, J.-H. 2D Materials for Skin-Mountable Electronic Devices. *Adv. Mater.* **2021**, *33* (47), 2005858.
- (8) Kireev, D.; Ameri, S. K.; Nederveld, A.; Kampfe, J.; Jang, H.; Lu, N.; Akinwande, D. Fabrication, Characterization and Applications of Graphene Electronic Tattoos. *Nat. Protoc.* **2021**, *16* (5), 2395–2417.
- (9) Pang, Y.; Yang, Z.; Yang, Y.; Ren, T.-L. Wearable Electronics Based on 2D Materials for Human Physiological Information Detection. *Small* **2020**, *16* (15), 1901124.
- (10) Kuila, T.; Bose, S.; Khanra, P.; Mishra, A. K.; Kim, N. H.; Lee, J. H. Recent Advances in Graphene-Based Biosensors. *Biosens. Bioelectron.* **2011**, *26* (12), 4637–4648.
- (11) Kireev, D.; Offenhäusser, A. Graphene & Two-Dimensional Devices for Bioelectronics and Neuroprosthetics. *2D Mater.* **2018**, *5* (4), 042004.
- (12) Hess, L. H.; Seifert, M.; Garrido, J. A. Graphene Transistors for Bioelectronics. *Proc. IEEE* **2013**, *101* (7), 1780–1792.
- (13) Wang, Q. H.; Kalantar-Zadeh, K.; Kis, A.; Coleman, J. N.; Strano, M. S. Electronics and Optoelectronics of Two-Dimensional Transition Metal Dichalcogenides. *Nat. Nanotechnol.* **2012**, *7* (11), 699–712.

- (14) Radisavljevic, B.; Radenovic, A.; Brivio, J.; Giacometti, V.; Kis, A. Single-Layer MoS₂ Transistors. *Nat. Nanotechnol.* **2011**, *6* (3), 147–150.
- (15) Manzeli, S.; Ovchinnikov, D.; Pasquier, D.; Yazyev, O. V.; Kis, A. 2D Transition Metal Dichalcogenides. *Nat. Rev. Mater.* **2017**, *2* (8), 17033.
- (16) Ye, M.; Zhang, D.; Yap, Y. K. Recent Advances in Electronic and Optoelectronic Devices Based on Two-Dimensional Transition Metal Dichalcogenides. *Electronics* **2017**, *6* (2), 43.
- (17) Chen, Y.; Wang, Y.; Wang, Z.; Gu, Y.; Ye, Y.; Chai, X.; Ye, J.; Chen, Y.; Xie, R.; Zhou, Y.; Hu, Z.; Li, Q.; Zhang, L.; Wang, F.; Wang, P.; Miao, J.; Wang, J.; Chen, X.; Lu, W.; Zhou, P.; Hu, W. Unipolar Barrier Photodetectors Based on van Der Waals Heterostructures. *Nat. Electron.* **2021**, *4* (5), 357–363.
- (18) Ippolito, S.; Kelly, A. G.; Furlan de Oliveira, R.; Stoeckel, M.-A.; Iglesias, D.; Roy, A.; Downing, C.; Bian, Z.; Lombardi, L.; Samad, Y. A.; Nicolosi, V.; Ferrari, A. C.; Coleman, J. N.; Samori, P. Covalently Interconnected Transition Metal Dichalcogenide Networks via Defect Engineering for High-Performance Electronic Devices. *Nat. Nanotechnol.* **2021**, *16* (5), 592–598.
- (19) Kelly, A. G.; Hallam, T.; Backes, C.; Harvey, A.; Esmaeily, A. S.; Godwin, I.; Coelho, J.; Nicolosi, V.; Lauth, J.; Kulkarni, A.; Kinge, S.; Siebbeles, L. D. A.; Duesberg, G. S.; Coleman, J. N. All-Printed Thin-Film Transistors from Networks of Liquid-Exfoliated Nanosheets. *Science* **2017**, *356* (6333), 69–73.
- (20) Grillo, A.; Parvez, K.; Wang, J.; Peng, Z.; Pelella, A.; Dun, C.; Urban, J. J.; Di Bartolomeo, A.; Casiraghi, C. All 2D Material Printed Diodes and Circuits on Paper for Sustainable Electronics. *ACS Nano* **2025**, *19* (34), 30833–30843.
- (21) Yang, P.; Zhu, L.; Zhou, F.; Zhang, Y. Wafer-Scale Uniform Synthesis of 2D Transition Metal Dichalcogenides Single Crystals via Chemical Vapor Deposition. *Acc. Mater. Res.* **2022**, *3* (2), 161–174.
- (22) Hu, J.; Zhou, F.; Wang, J.; Cui, F.; Quan, W.; Zhang, Y. Chemical Vapor Deposition Syntheses of Wafer-Scale 2D Transition Metal Dichalcogenide Films toward Next-Generation Integrated Circuits Related Applications. *Adv. Funct. Mater.* **2023**, *33* (40), 2303520.
- (23) Piva, N.; Greco, F.; Garbugli, M.; Iacchetti, A.; Mattoli, V.; Caironi, M. Tattoo-Like Transferable Hole Selective Electrodes for Highly Efficient, Solution-Processed Organic Indoor Photovoltaics. *Adv. Electron. Mater.* **2018**, *4* (10), 1700325.
- (24) Ferrari, L. M.; Keller, K.; Burtscher, B.; Greco, F. Temporary Tattoo as Unconventional Substrate for Conformable and Transferable Electronics on Skin and Beyond. *Multifunct. Mater.* **2020**, *3* (3), 032003.
- (25) Ferrari, L. M.; Ismailov, U.; Badier, J.-M.; Greco, F.; Ismailova, E. Conducting Polymer Tattoo Electrodes in Clinical Electro- and Magneto-Encephalography. *Npj Flex. Electron.* **2020**, *4* (1), 4.
- (26) Ferrari, L. M.; Sudha, S.; Tarantino, S.; Esposti, R.; Bolzoni, F.; Cavallari, P.; Cipriani, C.; Mattoli, V.; Greco, F. Ultraconformable Temporary Tattoo Electrodes for Electrophysiology. *Adv. Sci.* **2018**, *5* (3), 1700771.
- (27) Burtscher, B.; Leising, G.; Greco, F. Temporary Tattoo Approach for a Transferable Printed Organic Photodiode. *ACS Appl. Electron. Mater.* **2021**, *3* (6), 2652–2660.
- (28) Bonacchini, G. E.; Bossio, C.; Greco, F.; Mattoli, V.; Kim, Y.-H.; Lanzani, G.; Caironi, M. Tattoo-Paper Transfer as a Versatile Platform for All-Printed Organic Edible Electronics. *Adv. Mater.* **2018**, *30* (14), 1706091.
- (29) Sozen, Y.; Riquelme, J. J.; Xie, Y.; Munuera, C.; Castellanos-Gomez, A. High-Throughput Mechanical Exfoliation for Low-Cost Production of van Der Waals Nanosheets. *Small Methods* **2023**, *7* (10), 2300326.
- (30) Yang, V.; Bastiani, N. P. D. Inkjet Printable Waterslide Transferable Media. US6623817B1, September 23, 2003. <https://patents.google.com/patent/US6623817B1/en> (accessed 2025–12–11).
- (31) Dong-wook, J.. Manufacturing Method of Inkjet Photo Transfer Paper for Digital Photo Printing and Transfer Printing Method Using the Same. KR100790113B1, December 31, 2007. <https://patents.google.com/patent/KR100790113B1/en> (accessed 2025–12–11).
- (32) Sozen, Y.; Pucher, T.; Kesavan, B. P.; Jimenez-Arevalo, N.; Hernandez-Ruiz, J.; Sofer, Z.; Munuera, C.; Riquelme, J. J.; Castellanos-Gomez, A. Wafer-Scale Films of Two-Dimensional Materials via Roll-to-Roll Mechanical Exfoliation. arXiv November 10, 2025.
- (33) Chakraborty, B.; Matte, H. S. S. R.; Sood, A. K.; Rao, C. N. R. Layer-Dependent Resonant Raman Scattering of a Few Layer MoS₂. *J. Raman Spectrosc.* **2013**, *44* (1), 92–96.
- (34) Li, H.; Zhang, Q.; Yap, C. C. R.; Tay, B. K.; Edwin, T. H. T.; Olivier, A.; Baillargeat, D. From Bulk to Monolayer MoS₂: Evolution of Raman Scattering. *Adv. Funct. Mater.* **2012**, *22* (7), 1385–1390.
- (35) Ogilvie, S. P.; Large, M. J.; Wood, H. J.; Amorim Graf, A.; Lee, F.; Salvage, J. P.; King, A. A. K.; Dalton, A. B. Size Selection and Thin-Film Assembly of MoS₂ Elucidates Thousandfold Conductivity Enhancement in Few-Layer Nanosheet Networks. *Nanoscale* **2022**, *14* (2), 320–324.
- (36) Kalosi, A.; Demydenko, M.; Bodik, M.; Hagara, J.; Kotlar, M.; Kostiuik, D.; Halahovets, Y.; Vegso, K.; Marin Roldan, A.; Maurya, G. S.; Angus, M.; Veis, P.; Jergel, M.; Majkova, E.; Siffalovic, P. Tailored Langmuir–Schaefer Deposition of Few-Layer MoS₂ Nanosheet Films for Electronic Applications. *Langmuir* **2019**, *35* (30), 9802–9808.
- (37) Mazaheri, A.; Lee, M.; van der Zant, H. S. J.; Frisenda, R.; Castellanos-Gomez, A. MoS₂-on-Paper Optoelectronics: Drawing Photodetectors with van Der Waals Semiconductors beyond Graphite. *Nanoscale* **2020**, *12* (37), 19068–19074.
- (38) Pecunia, V.; Anthopoulos, T. D.; Armin, A.; Bouthinon, B.; Caironi, M.; Castellanos-Gomez, A.; Chen, Y.; Cho, K.; Clegg, C.; Fang, X.; Fendel, P.; Fowler, B.; Gelinck, G.; Gottlob, H.; Guyot-Sionnest, P.; Hannebauer, R.; Hernandez-Sosa, G.; Hersam, M. C.; Hirsch, L.; Ho, J. C.; Isikgor, F. H.; Joimel, J.; Kim, H. J.; Konstantatos, G.; Labram, J.; Lemme, M. C.; Leo, K.; Lhuillier, E.; Lidorikis, E.; Loi, M. A.; Malinowski, P. E.; Merken, P.; Mueller, T.; Nasrollahi, B.; Natali, D.; Ng, T. N.; Nguyen, T.-Q.; Park, S. K.; Peng, L.-M.; Samori, P.; Sargent, E. H.; Shen, L.; Shishido, S.; Shorubalko, I.; Sonar, P.; Stranks, S. D.; Tedde, S. F.; Vandewal, K.; Verhaegen, M.; Walia, S.; Yan, F.; Yokota, T.; Zhang, F. Guidelines for Accurate Evaluation of Photodetectors Based on Emerging Semiconductor Technologies. *Nat. Photonics* **2025**, *19* (11), 1178–1188.
- (39) Sessa, A.; De Stefano, S.; Durante, O.; Pelella, A.; Aldrigo, M.; Parvulescu, C.; Dinescu, A.; Kuo, C.-N.; Lue, C. S.; Dadiani, T.; D'Olimpio, G.; Faella, E.; Politano, A.; Passacantando, M.; Di Bartolomeo, A. Synaptic Behavior in SnSe₂ Field-Effect Transistors Induced by Surface Oxide and Trap Dynamics. *Adv. Electron. Mater.* **2026**, *12* (5), No. e00734.
- (40) Pucher, T.; Bastante, P.; Sánchez Viso, E.; Castellanos-Gomez, A. Low-Cost Shadow Mask Fabrication for Nanoelectronics. *Nanomanufacturing* **2023**, *3* (3), 347–355.
- (41) Daus, A.; Jaikissoon, M.; Khan, A. I.; Kumar, A.; Grady, R. W.; Saraswat, K. C.; Pop, E. Fast-Response Flexible Temperature Sensors with Atomically Thin Molybdenum Disulfide. *Nano Lett.* **2022**, *22* (15), 6135–6140.
- (42) Neilson, J.; Caffrey, E.; Cassidy, O.; Gabbett, C.; Synnatschke, K.; Schneider, E.; Munuera, J. M.; Carey, T.; Rimmer, M.; Sofer, Z.; Maultzsch, J.; Haigh, S. J.; Coleman, J. N. Production of Ultrathin and High-Quality Nanosheet Networks via Layer-by-Layer Assembly at Liquid–Liquid Interfaces. *ACS Nano* **2024**, *18* (47), 32589–32601.
- (43) Lee, K. H.; Kang, M. S.; Zhang, S.; Gu, Y.; Lodge, T. P.; Frisbie, C. D. “Cut and Stick” Rubbery Ion Gels as High Capacitance Gate Dielectrics. *Adv. Mater.* **2012**, *24* (32), 4457–4462.
- (44) Zhang, S.; Lee, K. H.; Frisbie, C. D.; Lodge, T. P. Ionic Conductivity, Capacitance, and Viscoelastic Properties of Block Copolymer-Based Ion Gels. *Macromolecules* **2011**, *44* (4), 940–949.
- (45) Lee, K. H.; Zhang, S.; Lodge, T. P.; Frisbie, C. D. Electrical Impedance of Spin-Coatable Ion Gel Films. *J. Phys. Chem. B* **2011**, *115* (13), 3315–3321.
- (46) Gao, X.; Yin, J.; Bian, G.; Liu, H.-Y.; Wang, C.-P.; Pang, X.-X.; Zhu, J. High-Mobility Patternable MoS₂ Percolating Nanofilms. *Nano Res.* **2021**, *14* (7), 2255–2263.
- (47) Chen, M.; Cui, D.; Wang, N.; Weng, S.; Zhao, Z.; Tian, F.; Gao, X.; He, K.; Chiang, C.-T.; Albawardi, S.; Alsaggaf, S.; Aljalham, G.;

Amer, M. R.; Zhou, C. Inkjet-Printed MoS₂ Nanoplates on Flexible Substrates for High-Performance Field Effect Transistors and Gas Sensing Applications. *ACS Appl. Nano Mater.* **2023**, *6* (5), 3236–3244.

(48) Song, O.; Rhee, D.; Kim, J.; Jeon, Y.; Mazánek, V.; Söll, A.; Kwon, Y. A.; Cho, J. H.; Kim, Y.-H.; Sofer, Z.; Kang, J. All Inkjet-Printed Electronics Based on Electrochemically Exfoliated Two-Dimensional Metal, Semiconductor, and Dielectric. *npj 2D Mater. Appl.* **2022**, *6* (1), 64.

(49) Carey, T.; Cassidy, O.; Synnatschke, K.; Caffrey, E.; Garcia, J.; Liu, S.; Kaur, H.; Kelly, A. G.; Munuera, J.; Gabbett, C.; O'Suilleabhain, D.; Coleman, J. N. High-Mobility Flexible Transistors with Low-Temperature Solution-Processed Tungsten Dichalcogenides. *ACS Nano* **2023**, *17* (3), 2912–2922.

(50) Kwon, Y. A.; Kim, J.; Jo, S. B.; Roe, D. G.; Rhee, D.; Song, Y.; Kang, B.; Kim, D.; Kim, J.; Kim, D. W.; Kang, M. S.; Kang, J.; Cho, J. H. Wafer-Scale Transistor Arrays Fabricated Using Slot-Die Printing of Molybdenum Disulfide and Sodium-Embedded Alumina. *Nat. Electron.* **2023**, *6* (6), 443–450.

(51) Kelly, A. G.; O'Suilleabhain, D.; Gabbett, C.; Coleman, J. N. The Electrical Conductivity of Solution-Processed Nanosheet Networks. *Nat. Rev. Mater.* **2022**, *7* (3), 217–234.

(52) Ippolito, S.; Urban, F.; Zheng, W.; Mazzarisi, O.; Valentini, C.; Kelly, A. G.; Gali, S. M.; Bonn, M.; Beljonne, D.; Corberi, F.; Coleman, J. N.; Wang, H. I.; Samori, P. Unveiling Charge-Transport Mechanisms in Electronic Devices Based on Defect-Engineered MoS₂ Covalent Networks. *Adv. Mater.* **2023**, *35* (15), 2211157.

(53) Chen, X.; Bartlam, C.; Lloret, V.; Moses Badlyan, N.; Wolff, S.; Gillen, R.; Stimpel-Lindner, T.; Maultzsch, J.; Duesberg, G. S.; Knirsch, K. C.; Hirsch, A. Covalent Bisfunctionalization of Two-Dimensional Molybdenum Disulfide. *Angew. Chem., Int. Ed.* **2021**, *60* (24), 13484–13492.

(54) Park, Y. J.; Sharma, B. K.; Shinde, S. M.; Kim, M.-S.; Jang, B.; Kim, J.-H.; Ahn, J.-H. All MoS₂-Based Large Area, Skin-Attachable Active-Matrix Tactile Sensor. *ACS Nano* **2019**, *13* (3), 3023–3030.

(55) Daus, A.; Jaikissoon, M.; Khan, A. I.; Kumar, A.; Grady, R. W.; Saraswat, K. C.; Pop, E. Fast-Response Flexible Temperature Sensors with Atomically Thin Molybdenum Disulfide. *Nano Lett.* **2022**, *22* (15), 6135–6140.

(56) Zhong, D.; Wu, C.; Jiang, Y.; Yuan, Y.; Kim, M.; Nishio, Y.; Shih, C.-C.; Wang, W.; Lai, J.-C.; Ji, X.; Gao, T. Z.; Wang, Y.-X.; Xu, C.; Zheng, Y.; Yu, Z.; Gong, H.; Matsuhisa, N.; Zhao, C.; Lei, Y.; Liu, D.; Zhang, S.; Ochiai, Y.; Liu, S.; Wei, S.; Tok, J. B.-H.; Bao, Z. High-Speed and Large-Scale Intrinsically Stretchable Integrated Circuits. *Nature* **2024**, *627* (8003), 313–320.

(57) Quereda, J.; Zhao, Q.; Diez, E.; Frisenda, R.; Castellanos-Gomez, A. Fiber-Coupled Light-Emitting Diodes (LEDs) as Safe and Convenient Light Sources for the Characterization of Optoelectronic Devices. *Open Res. Eur.* **2021**, *1*, 98.



CAS INSIGHTS™
**EXPLORE THE INNOVATIONS
SHAPING TOMORROW**

Discover the latest scientific research and trends with CAS Insights. Subscribe for email updates on new articles, reports, and webinars at the intersection of science and innovation.

Subscribe today

CAS
Advances of the
American Chemical Society